

# **Anisotropic magneto-thermoelectric power of ferromagnetic** thin films

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#### Abstract

Magnetothermoelectric power (MTEP) in metallic ferromagnetic films of Ni<sub>80</sub>Fe<sub>20</sub> (Permalloy; Py), Co and CrO<sub>2</sub> was measured at temperatures in the range of 100K to 400K. In Py and Co both the anisotropic magnetoresistance (AMR) and MTEP show a relative change in resistance and thermopower of the order of 0.15%. The CrO<sub>2</sub> films were grown on TiO<sub>2</sub> and on sapphire. Where MTEP behavior at low fields shows peaks similar to the AMR, with variations up to 0.9%. With increasing field both the MR and the MTEP variations keeps growing, with MTEP showing relative changes of 1.5% with the thermal gradient along the b-axis and even 20% with the gradient along the c-axis, with an intermediate value of 3% for the film on sapphire. It appears that the low-field effects are due to the magnetic domain state, and the high-field effects are intrinsic to the electronic structure of CrO<sub>2</sub>.





Semi classical, Mott's formula of Seebeck coefficient, ratio of derivative of density of states and the density of states at Fermi level

$$S = -eL_oT \frac{N'(\mathcal{E}_F)}{N(\mathcal{E}_F)}$$

S in ferromagnetic material, electron magnon scattering, spin dependent S with opposite sign

$$S = -eL_oT\left\{\frac{N'\uparrow(\mathcal{E}_F)}{N(\mathcal{E}_F)} - \frac{N'\downarrow(\mathcal{E}_F)}{N(\mathcal{E}_F)}\right\} = (S\uparrow -S\downarrow), N = N\uparrow +N\downarrow$$

Influence of magnetic domain structure on thermoelectric power TEP?

Temperature dependent Seebeck coefficient was measured via dynamic technique in the Py films were deposited in UHV magnetron sputtering system with background pressure 10<sup>-9</sup> mbar, Co films were deposited in Z-400 background presure of 10<sup>-6</sup> mbar.  $CrO_2$  thin films were deposited on  $TiO_2$  and sapphire substrates utilizing the Chemical Vapor Deposition technique at atmospheric pressure

Figure 2. AMF images of CrO<sub>2</sub> films deposited on (a) thin Sapphire (b) TiO<sub>2</sub> substrates. Films grow in the form of rectangular grains.

Figure 1. Home build thermal transport sample holder, based on PPMS puck.





Results



**MTEP** measurements



Figure 5, (a) MTEP along c-axis of  $CrO_2$  thin films deposited on  $TiO_2$ substrate, (b) MTEP along b-axis. (c) and (d) are illustrating AMR signal along c-axis and b-axis respectively.

Both R and S are increasing in domain state but opposite for Py.





Figure 4, (a) S(T) for  $CrO_2$  films deposited on  $TiO_2$  and Sapphire substrates, (b) R(T) for  $CrO_2$  for both of these films, (c) MTEP for CrO<sub>2</sub> deposited on Sapphire (d) and corresponding AMR.

Figure 6. MTEP with relative change of 20% along c-axis of  $CrO_2$  thin films deposited on  $TiO_2$  substrate.

Figure 7. MTEP for a multilayer ferromagnet sample Py(25nm)/CuNi(50nm), No peak from CuNi at  $T_{ave}=185>Tc$ of CuNi (a) and appearing at T<Tc (b).

## Conclusion

Magnetothermopower (MTEP) is strongly related with anisotropic magnetoresistance (AMR)

#### MTEP is increasing in the domain state.

Py and Co have opposite behaviour although their AMR have the same behaviour.

CrO2 shows a huge TEP variation when measured along its c-axis.'

S(T) of CrO2 thin films has anisotropic behavior.

### References

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